CFD modelling of erosion-corrosion of steels in aqueous environments: particle concentration effects on the regime boundaries.

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Abstract:

In this study, a new methodology is used to model the effects of particle concentration on the inner surfaces of a circular pipe 90° bend, assuming applied potential controlled aqueous slurry flow at room temperature. This enables the regimes of the component to be mapped according to the intensity of erosion and corrosion contributions. The results show that for a constant inlet particle concentration, it is shown how transitions between erosion-corrosion regimes are observed around the pipe. For increases in particle concentration, significant variation of the erosion-corrosion regimes are observed, with a reduction of the corrosion dominated regime. The results are interpreted in the terms of the changes in local erosion conditions along the component in the flowing environments. Typical results from the model are shown illustrating how this new mapping method can be used effectively to optimize process conditions and materials in such environments.

Key words: Erosion-Corrosion maps, CFD Modelling, mild steel.

1. NOTATION AND UNITS

| Latin letters: | 
|---|---|
| $A_{in}$ | Pipe inlet area. $[m^2]$ |
| $c$ | Particle concentration $[g \text{ cm}^{-3}]$ |
| $C_p$ | Specific heat capacity $[J \text{ kg}^{-1} \text{ K}^{-1}]$ |
| $d$ | Diameter of the pipe. $[m]$ |
| $E$ | Applied potential. $[V]$(SCE) |
| $e_n$ | Normal coeff. of restitution. |
| $F$ | Faraday constant. $(96485)\ [C \text{ mol}^{-1}]$ |
| $f_t$ | Numerical constant. $(0.025)$ |
| $H_s$ | Hardness of the substrate. $[Pa]$ |
| $h$ | Thickness of the oxide layer. $[m]$ |
| $h_o$ | Initial thickness of the oxide. $[m]$ |
| $I_p$ | Impact frequency. $[imp \text{ cm}^2 \text{ s}^{-1}]$ |
| $i_{anet}$ | Net anodic current density. $[A \text{ m}^{-2}]$ |
| $i_o$ | Exchange current density. $[A \text{ m}^{-2}]$ |
| $K_c$ | Pure corrosion rate. $[g \text{ m}^{-2} \text{ s}^{-1}]$ |
| $K_{ec}$ | Total wastage rate. $[g \text{ m}^{-2} \text{ s}^{-1}]$ |
| $K_e$ | Pure erosion rate. $[g \text{ m}^{-2} \text{ s}^{-1}]$ |
| $k$ | Material constant. $(0.699)$ |
| $k_2$ | Material constant. $(1.398)$ |
| $M_{r}$ | Amount of mass removal per $[g \text{ impact}]$ |
impact.

- \( m_p \):
  - Mass of impacting particle.
  - [kg]

- \( n_c \):
  - Strain hardening coefficient.
  - (0.3)

- \( RAM \):
  - Relative atomic mass of Fe.
  - (55.8) [g mol\(^{-1}\)]

- \( T_m \):
  - Melting point of the Fe.
  - (1808) [K]

- \( V_p \):
  - Particle velocity.
  - [m s\(^{-1}\)]

- \( \nu_f \):
  - Volume fraction of particles.

- \( W_C \):
  - Erosion by cutting.
  - [kg kg\(^{-1}\)]

- \( W_D \):
  - Erosion by deformation.
  - [kg kg\(^{-1}\)]

- \( z_m \):
  - Number of electrons.
  - (2)

**Greek letters:**

- \( \alpha_p \):
  - Impact angle
  - [deg]

- \( \lambda \):
  - Particle shape factor
  - (0.0)

- \( \mu \):
  - Frictional coefficient
  - (0.1)

- \( \mu_f \):
  - Critical friction coefficient
  - (Sundararajan)

- \( \pi \):
  - Pi ratio
  - (3.142)

- \( \rho_f \):
  - Density of the oxide layer.
  - (5240) [kg m\(^{-3}\)]

- \( \rho_a \):
  - Density of the particle
  - (2650) [kg m\(^{-3}\)]

**Subscripts:**

- \( ap \):
  - Applied potential.

- \( rev \):
  - Reversible equilibrium potential.

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### 2. INTRODUCTION

In studies of modelling erosion-corrosion, both finite element (FE) [1] and computational fluid dynamics (CFD) [2-4] techniques have been used to investigate mass loss of materials and to evaluate the effect of various parameters controlling the erosion and corrosion process. However, the majority of these investigations do not introduce a full definition for erosion-corrosion regimes resulting from the interaction between erosion and corrosion. These regimes are very useful in identifying the synergistic, additive, and antagonistic behaviour of this phenomenon in addition to identifying the mechanism of degradation [5].

An approach to identify the transition regimes for the erosion-corrosion process and also to monitor the effects of various parameters was developed by Stack et al. [5] as an extension of earlier work on erosion-corrosion mapping [6-8]. Although 2D mapping is capable of capturing the influence of any particular parameter on erosion-corrosion regime boundaries, it does not give any indication of the combined effect of all the parameters on these boundaries, nor the metal degradation in a 3D space.

A new methodology was introduced in previous work [9] to study the integrated effect of these parameters together (namely those related to the particle, fluid flow, and environment), on the boundaries of the erosion-corrosion regime and wastage maps by combining the concept of CFD with the erosion-corrosion mapping techniques. Such approach enables us to map the surfaces of any 3D component that are exposed to aqueous slurry flow. Furthermore, it provides a powerful predictive tool for estimating the predominance of various erosion-corrosion regimes.

In this study, a 3D mapping technique is used to investigate the effect of the particle concentration and on the construction of erosion-corrosion mechanism and wastage maps. The effects of particle concentration are evaluated on the regime boundaries. The results are discussed in terms of the applications of this technique to erosion-corrosion in slurry flows in addition to addressing some current limitations.

### 3. METHODOLOGY

A dilute slurry flow of water-alumina particles of size \( 10^{-3} [m] \) and four volume fractions i.e. 0.025, 0.05, 0.075, and 0.1 is ingested through a pipe bend inlet with bore diameter \((D)\) equal to 0.078 [m] and \((R D^{-1})\) ratio of 1.2.

The CFD simulation generated by FLUENT ver.6.3 [10] uses a finite element based finite volume method to solve the flow governing equations. Table 1 summarizes the equations used, operating and boundary conditions used in this study while Table 2 lists the mechanical and physical properties for the slurry and target material.

Validation of the erosion results is carried out in [9] by comparison to previous experimental work as described elsewhere [11], and is summarised in Table 3. The simulation validation was carried out
for a SS304L stainless steel alloy using Forder’s erosion model [12] as in the case study, and the results were simulated for mild steel using Sundararajan’s second model [13].

**Table 1. CFD modeling equations, operating and boundary conditions**

<table>
<thead>
<tr>
<th>Model parameter</th>
<th>Water</th>
<th>Sand</th>
<th>Particles</th>
</tr>
</thead>
<tbody>
<tr>
<td>Solver equations</td>
<td>Navier-Stokes</td>
<td>DPM</td>
<td></td>
</tr>
<tr>
<td>Turbulence model</td>
<td>Standard k-ε</td>
<td>Standard wall</td>
<td></td>
</tr>
<tr>
<td>Wall treatment</td>
<td>function- no slip</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Coupling</td>
<td>One way</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Operating cond.</td>
<td>ambient</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Inlet velocity $[m\ s^{-1}]$</td>
<td>3.0</td>
<td>3.0</td>
<td></td>
</tr>
</tbody>
</table>

**Table 2. Physical and mechanical properties for the slurry and target material**

<table>
<thead>
<tr>
<th></th>
<th>Fluid (water)</th>
<th>Particles (alumina)</th>
<th>Target (mild steel)</th>
</tr>
</thead>
<tbody>
<tr>
<td>density $[kg\ m^{-3}]$</td>
<td>998</td>
<td>2670</td>
<td>7850</td>
</tr>
<tr>
<td>particle size $[m]$</td>
<td>10^{-3}</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Flow rate $[kg\ s^{-1}]$</td>
<td>14.3</td>
<td>variable</td>
<td>1.398</td>
</tr>
</tbody>
</table>

**Table 3. Comparison between the current study and Experimental and simulation work of Wood et al. [11]**

<table>
<thead>
<tr>
<th></th>
<th></th>
<th></th>
<th></th>
</tr>
</thead>
<tbody>
<tr>
<td>Erosion rates $[mm^3\ imp^{-1}]$</td>
<td>2.2-5.5</td>
<td>5.5</td>
<td>5.45</td>
</tr>
</tbody>
</table>

**3.1. Erosion modelling**

The second model of Sundararajan is divided into two expressions; one accounts for the localised deformation at the impact point, whilst the other addresses the ductile cutting mechanism during the impact. The total erosion rate is the summation of these two mechanisms. The formulation is as follows [13]:

\[
\dot{W}_p = \frac{5.5 \times 10^{-2}}{(T_m - 436)^{0.75}} \left(2^{n_e} f_i V^2 \sin^2 \alpha (1 - e_n^2)\right) n_e C_p \tag{1}
\]

\[
\dot{W}_c = \frac{5.5 \times 10^{-2}}{(T_m - 436)^{0.75}} \left(\frac{\mu}{\mu_f} \left(2 - \frac{\mu}{\mu_f}\right) V^2 \cos^2 \alpha \right) \left(1 + \lambda\right)^{2 - n_e} n_e C_p \tag{2}
\]

where

\[
\mu_f = \frac{1}{(1 + \lambda)(1 + e) \tan \alpha} \tag{3}
\]

and $e_n$ is the normal restitution ratio [12] :

\[
e_n = 0.988 - 0.78 \alpha + 0.19 \alpha^2 - 0.024 \alpha^3 + 0.027 \alpha^4 \tag{4}
\]

For unit consistency, erosion rates should be converted to $[kg\ m^{-2}\ s^{-1}]$ in line with the calculated corrosion rates below.

**3.2. Corrosion rates**

**3.2.1. Active corrosion model**

In the active region, it is assumed that the total corrosion is estimated from knowledge of the dissolution current density from the Butler-Volmer equation:

\[
l_{avct} = l_o \left[\exp\left(\frac{\beta z_m Fr \Delta E}{R_o T}\right) - \exp\left(-\left(1 - \beta\right) z_m Fr \frac{\Delta E}{R_o T}\right)\right] \tag{5}
\]

where the over-potential $\Delta E$ is defined as:

\[
\Delta E = E_{aq} - E_{rev} \tag{6}
\]
The corrosion rate is therefore given by:

\[ K_c = K_{co} = \frac{RAM i_{anet}}{z_m F} \]  \hspace{1cm} (7)

### 3.2.2. Repassivation model

For the passivation mechanism, we assume that pure corrosion \( (K_{co}) \) rate is sufficiently low as to be negligible and the corrosion rate is the additive effect of corrosion enhanced by erosion \( (\Delta K_c) \) [14].

An expression for the corrosion rate in passivation conditions has been introduced [5] and is currently modified to include the effect of the oblique impact [9, 15].

\[ M_i = \pi k h \rho d_p^2 \left[ \frac{\rho_p (1-e_n^2)}{6H_s} \right]^{0.5} \left( V_p \sin \alpha_i \right) \]  \hspace{1cm} (8)

The constant \( k \) is defined as the mass ratio between the metal and the oxide created during the corrosion reaction, multiplied by the number of moles of metal involved in the reaction and is related to \( k_2 \) in [14] by definition as \( (k = k_2/2) \).

The thickness of the passive layer \( h \) can be assumed to vary with the potential difference and may be given from [16] assuming a linear relationship between the over-potential and passive layer thickness:

\[ h = h_o + 3 \times 10^{-9} \left( E_{ap} - E_{pav} \right) \]  \hspace{1cm} (9)

and: \( h_o = 1 \times 10^{-9} [m] \).

The unit given by equation (8) is \([kg \text{ impact}^{-1}]\). To convert to \([kg \text{ m}^2 \text{ s}^{-1}]\), it is multiplied by particle impact flux as outlined in [14]. This can be varied according to the application under investigation. For example, if the flow is homogeneous then particle impact frequency may be used and is given as [17]:

\[ K_c = \Delta K_c = \frac{M_i c V_p \sin \alpha}{m_p} \]  \hspace{1cm} (10)

### 3.2.3. Erosion-corrosion boundaries

The regime boundaries are determined in terms of the ratio \( K_c/K_e \) as

\[ \frac{K_c}{K_e} < 0.1 \]  \hspace{1cm} (Erosion dominated) \hspace{1cm} (11)

\[ 0.1 \leq \frac{K_c}{K_e} < 1 \]  \hspace{1cm} (Erosion-Corrosion dominated) \hspace{1cm} (12)

\[ 1 \leq \frac{K_c}{K_e} < 10 \]  \hspace{1cm} (Corrosion-Erosion dominated) \hspace{1cm} (13)

\[ \frac{K_c}{K_e} \geq 10 \]  \hspace{1cm} (Corrosion dominated) \hspace{1cm} (14)

The transition boundaries for the wastage maps are given as follows:

\[ K_{ec} < 1 \text{ [mm year}^{-1}\text{]} \]  \hspace{1cm} (low wastage) \hspace{1cm} (15)

\[ 1 \leq K_{ec} < 10 \text{ [mm year}^{-1}\text{]} \]  \hspace{1cm} (medium wastage) \hspace{1cm} (16)

\[ K_{ec} \geq 10 \text{ [mm year}^{-1}\text{]} \]  \hspace{1cm} (high wastage) \hspace{1cm} (17)

### 4. RESULTS

Figure 1 illustrates the Pourbaix diagram for Fe at \( T=298 \text{ [K]} \) and identifies the possible corrosion mechanisms (i.e. immunity, dissolution and passivation mechanisms) for various values of temperature, pH and applied potential.
To investigate the particle concentration effect, a homogeneous particle distribution in the pipe inlet is assumed and the particle concentration is thus related to the particle mass flow rate according to the relation:

$$\dot{m}_p = \alpha_p \rho_p V_p A_{in}$$  \hspace{1cm} (18)

The simulation was thus run at various particle mass flow rates; namely 0.957, 1.9169, 2.87 and 3.8276 [kg s$^{-1}$] which are equivalent to particle inlet volume fractions of 0.025, 0.05, 0.075 and 0.0909 respectively.

Figure 2 shows the regime maps at the inner surfaces of the pipe bend at pH=7 and applied potential $E= -0.6$ [V] (SCE), illustrating the changes in erosion-corrosion regimes with increasing the particle volume fraction. As the particle volume fraction increases, the erosion rates increases linearly and as a result, the erosion-dissolution and erosion dominated regimes are enlarged. In addition, the dissolution dominated regime reduces with increasing particle volume fraction. This indicates the importance of the volume fraction (or particle concentration) as a factor affecting the erosion-corrosion regimes.

Figure 3 illustrates the wastage maps for pH=7 and at applied potential $E= -0.6$ [V] (SCE) for the same particle mass flow rates listed above as in figure 2. Again the high wastage regime area (red colour region) is increased as the particle mass flow rate increases.
5. DISCUSSION

The results indicate that particle volume fraction effects play a significant role in changing the erosion-corrosion regime in 3-D, Figs. 2-3. In the 3-D case, various regimes are observed over the surface, Fig. 2. With increasing volume fraction, the erosion-dominated regime is enhanced. This is due to the increase in erosion rate with increasing particle concentration, shifting the corrosion affected regimes towards those dominated by erosion.

The above trends in the variation of erosion-corrosion regimes on the pipe bend are also observed in earlier work by the current authors [9]. In this work, the significant changes in particle concentration and velocity for the component are identified, resulting in a variation of erosion intensity on the surface. In addition, the local impact angle also changes. Hence, the erosion-corrosion regime variation on the component can be attributed to these factors.

Nonetheless, it should be noted that the variation of the erosion-corrosion regimes only occurs at the pipe bend only and the remaining parts (i.e. entrance and exit straight pipes) exhibit no erosion-corrosion regime change (results not shown). This indicates that the erosion rates at these regions remain unaffected during the simulation. For the wastage maps, the pipe exit exhibits a minor variation from the medium to the high wastage regime as the particle concentration increases. Hence, the erosion-corrosion mapping technique is a potentially sensitive method of detecting any change in the erosion-corrosion behavior in such environments where changes in component geometry are anticipated.

For the present study, additive erosion-corrosion behaviour is assumed i.e. erosion enhances corrosion and the effect of corrosion on erosion is negligible. The latter effect, the so-called “synergistic/antagonistic” interaction has been attributed to a number of possible mechanisms relating to the material microstructure [18, 19] and it is acknowledged that this is a simplistic assumption. Hence, initial work on mapping the component for the materials above has made such assumptions but future work will address such issues in more detail by assessing materials with composite structures.

Hence, 3D erosion-corrosion maps provide a useful tool for predicting and identification the erosion-corrosion regimes and wastage regions for in-service conditions. The results have indicated significant changes in regimes along the geometry of the component and as a function of increasing particle volume concentration in the pipe. Further work will be to model the effects of other parameters such as oxygen concentration and temperature in addition to extending the range of materials in the model and the synergism/anatagonism between the processes.

6. CONCLUSIONS

- CFD methods, involving fluid dynamics and multiphase flow parameters, have been used to model the erosion-corrosion behaviour of Fe at range of particle concentration values for a 3-D space.
The results indicate changes in erosion-corrosion regimes over the component geometry. Increases in particle concentration have a significant effect on the boundaries on the erosion-corrosion regime maps, reducing the corrosion affected regimes in favour of those dominated by erosion.

7. REFERENCES


