Surface Finish Control by Electrochemical Polishing in Stainless Steel 316 Pipes
A. A. Gomez-Gallegos\textsuperscript{a}, F. Mill\textsuperscript{a}, A. R. Mount\textsuperscript{b}
\textsuperscript{a} School of Engineering, The University of Edinburgh, Edinburgh, EH9 3JL, United Kingdom
\textsuperscript{b} School of Chemistry, The University of Edinburgh, Edinburgh, EH9 3JJ, United Kingdom

*Corresponding author: a.gomez@ed.ac.uk

Abstract
Electrochemical machining (ECM) is a non-conventional machining process which is based on the localised anodic dissolution of any conductive material. One of the main applications of ECM is the polishing of materials with enhanced characteristics, such as high strength, heat-resistance or corrosion-resistance, i.e. electrochemical polishing. The present work presents an evaluation of the parameters involved in the ECM of Stainless Steel 316 (SS316) with the objective of predicting the resulting surface finish on the sample. The interest of studying ECM on SS316 resides on the fact that a repeatable surface finish is not easily achieved. ECM experimental tests on SS316 pipes of 1.5" (0.0381 m) diameter were conducted by varying machining parameters such as voltage, interelectrode gap, electrolyte inlet temperature, and electrolyte flow rate. The surface finish of the samples was then evaluated in order to find the significance of each of these parameters on the surface quality of the end product. Results showed that overvoltage, which is dependent on the interelectrode gap and the electrolyte temperature, is one of the main parameters affecting the surface finish; additionally there is a strong relationship between the resulting surface finish and the electrolyte flow. The interelectrode gap and inlet electrolyte temperature also affect the resulting surface finish but their influence was not so evident in this work. Finally, the variation of the electrolyte temperature during the process was found to have a great impact on the uniformity of the surface finish along the sample.

Keywords: electrochemical machining, stainless steel 316, surface finish.

Introduction
ECM of metals with special characteristics, such as enhanced strength, heat or corrosion resistance, is a manufacturing option to produce products that could be difficult or impossible to get with conventional manufacturing processes. ECM allows manufacturers to shape any conductive material without affecting the properties of the tool or the workpiece. In addition, ECM can generate a high quality surface finish at the workpiece.

ECM consists of an electric circuit formed by the tool and the workpiece connected to an external electrical source. The electrodes are submerged in an electrolyte bath that closes the circuit. When current passes through the circuit, a localised anodic dissolution occurs at the workpiece. This results more or less in the negative shape of the tool profile \cite{1}. The electrolyte is pumped through the interelectrode gap dragging the dissolved material away and cooling down the electrodes. Unfortunately, the ECM process is difficult to
predict due to the wide variety of physical phenomena involved and the lack of sufficient quantitative and qualitative data that can be used to develop an accurate simulation model [2, 3].

One of the main applications of ECM is the polishing of materials, i.e. electrochemical polishing. Several studies of electrochemical polishing can be found in the published research [4-10], and studies on applying electrochemical polishing for the manufacture of biomedical implants [11], solar cells [12] and electrodes for photoelectron guns [13], are still under development. However, electrochemical polishing is a process that regularly generates non-repeatable results, e.g. the application of the process on stainless steels (SS) typically generates widely variable surface finish. Therefore, in many of these studies, special attention has been given to ECM on steels with high chromium content. Iron-chromium alloys, such as stainless steels, have wide applications in industry due to their characteristic behaviour (oxidation resistance). The chromium in SS induces the formation of a protective film of oxide on the material surface that prevents further corrosion [8]. However, this oxide film also modifies the ECM at the surface of the material; it has low electrical conductivity and prevents the workpiece from making direct contact with the electrolyte. Hence, normal anodic dissolution cannot be implemented without the breakdown of the oxide film. Partial breakdown of this film often occurs, which causes pitting on the surface [4, 8] or a non-uniform surface finish [6, 7, 14].

The surface finish of the samples results from the specular or non-specular reflection of light from the crystal faces that have been electrochemically dissolved at different rates during the ECM process [1, 15]. An electrochemically polished surface is usually associated with the random removal of atoms from the anode (workpiece) [15]. Datta and Landolt [16] observed that an active dissolution of material at low current density leads to surface etching, and a transpassive dissolution at high current density leads to surface brightening. This was also noted by Lee [5] and McGeough [1]; in their studies when the current density was raised, the surface finish on the workpiece became smoother. Lozano-Morales [14] found the same behaviour with ECM in Niobium samples. Elsewhere, Wagner, T. and Wang, et. al. [17, 18] gave a plausible explanation of the effect of the electrolyte flow rate on the surface finish on SS. They argue that during ECM, the electric current breaks the oxide film and local electrolyte flow turbulences would ideally remove the film particles (oxides, chromium carbides and reaction products). If the turbulence is not sufficient for removing these lose particles, Fe\textsuperscript{3+}- diffusion through the surface layer is possible, and the current density efficiency decreases drastically, thus affecting the surface finish.

In other works, Lee [5] found that when electropolishing stainless steel 316L, better electropolishing results were obtained at a temperature close to 68 °C than at 30 °C. Later on, Deconinck [2] demonstrated how the electrochemical reactions rates depend strongly on the electrolyte temperature, which in turn depends on the electrolyte flow rate, interelectrode gap and potential applied.

Mount et al. [6-8] built a segmented tool for the further analysis of the ECM process when applied on SS. Their results described a change in the surface finish of the samples along the electrolyte flow. It was found that this special behaviour is due to the change of the dissolution valence of the alloy. Their results showed
that for high dissolution valences, \( z = 3.5 \pm 0.1 \), the resulting surface was reflective and bright. This surface fish is characteristic of iron and chromium dissolving in their high valence states (as Fe(III), \( z_{Fe} = 3 \), and Cr(VI), \( z_{Cr} = 6 \)); however, if \( z \) was lower, \( 2.5 \pm 0.1 \), the surface finish was passivated. This surface finish is also characteristic of iron and chromium dissolving but now in their low valence states (as Fe(II), \( z_{Fe} = 2 \), and Cr(III), \( z_{Cr} = 3 \)) [8]. These results are in agreement with the study of Lohrengel et al. [3] and Murkherjee, et al. [19].

From the above, we can conclude that it is essential to understand how the ECM parameters affect the surface finish. The effective removal of this film is governed by a combination of metal-electrolyte-machining parameters. Hence, in the present work, the ECM machining parameters, gap, voltage, flow rate and inlet electrolyte temperature, are modified in order to evaluate their role on the achievement of the expected surface finish and a homogeneous breakdown of the oxide film. To the best of our knowledge, this is the first time that the inner surface finish of stainless steel commercial pipes has been enhanced by electrochemical polishing to tailor it for industrial applications and large scale production.

**Experimental method**

**Sample preparation**

The pipes machined were commercial stainless steel 316 (SS316) pipes of 0.17 m length and 0.0381 m diameter, which were manufactured by rolling and welding. The surface finish of the pipe prior to processing was dark and opaque and its quality was uniform along the pipe. Welding left behind a weld-flash at the interior face of the pipe. The exterior of the pipe was not treated.

**Electrochemical (ECM) setup**
**Figure 1. Schematic of the electrochemical machining system.**

Figure 1 shows the ECM array consisting of a cylindrical solid tool and a pipe (workpiece) placed vertically and concentric with each other on the bedplate of the ECM machine. The tool was held by the machine head and the workpiece was fixed to the bedplate by a non-conductive clamp. The electrolyte flowed from the top and between the pipe and the tool, and the setup was positioned in such a fashion that allows the electrolyte to exit the array. The temperature of the electrolyte was measured in situ at the entrance of the pipe using a digital thermometer.

**Test parameters**

The tool had the similar dimensions as the pipe. Its diameter offset from of the interior diameter of the pipe (workpiece) but undersized radially by 2, 4 or 8 mm. Electrical clips were connected to the tool-workpiece array providing a constant voltage of 18, 24 and 36 Volts (possible voltage losses in the system were not considered). The experimental parameters are summarised in Table 1.

<table>
<thead>
<tr>
<th>Name</th>
<th>Value</th>
</tr>
</thead>
<tbody>
<tr>
<td>$Y$</td>
<td>Interelectrode gap 2, 4, 8 mm</td>
</tr>
<tr>
<td>$V_1$</td>
<td>Voltage 18, 24, 36 V</td>
</tr>
<tr>
<td>$Q$</td>
<td>Electrolyte flow rate $1.7\times10^{-4}$, $4.2\times10^{-4}$, $6.7\times10^{-4}$, $10\times10^{-4}$ m$^3$/s</td>
</tr>
<tr>
<td>$T_e$</td>
<td>Inlet electrolyte temperature 7, 15.3 °C</td>
</tr>
</tbody>
</table>

The electrolyte used was Sodium Nitrate (NaNO$_3$) with specific gravity (S.G.) of 1.15. Electrolyte flow rate was recirculated and set at $1.7\times10^{-4}$, $4.2\times10^{-4}$, $6.7\times10^{-4}$, $10\times10^{-4}$ m$^3$/s (10, 25, 40 and 60 L/min). The electrolyte was stored in a pool at room temperature, hence the inlet electrolyte temperature was considered constant. However, the machining was developed on two different days, so the inlet temperature changed from 7 to 15.3 °C. Each test lasted for 10 seconds.

**Results and discussion**

Table 2 summarises the resulting surface finish qualities of the SS samples after ECM. The samples were divided in four categories according to their surface finish quality: (a) passivated entrance – reflective and bright exit; (b) reflective and bright; (c) reflective and dark; and (d) passivated.
**Table 2. Surface finish classification of the SS316 pipes machined by ECM.**

<table>
<thead>
<tr>
<th>(a) Passivated entrance – reflective and bright exit</th>
<th>(b) Reflective and bright</th>
</tr>
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<tbody>
<tr>
<td><img src="image1" alt="Passivated Entrance" /> <img src="image2" alt="Reflective and Bright Exit" /></td>
<td></td>
</tr>
<tr>
<td><img src="image3" alt="Passivated Entrance" /> <img src="image4" alt="Passivated Exit" /></td>
<td></td>
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**Passivated entrance – reflective and bright exit**

Table 1(a) presents a typical sample with a passivated surface finish at the entrance and a reflective and bright surface finish at the exit. An average roughness in the top part of the sample (passivated entrance) of 2.303 μm was measured with a Mitutoyo® profilometer. A small ring of passivated surface at the exit of the pipe is visible; this passivation is due to the accumulation of the electrolyte in this area forming a small pool before it can exit the array. The weld-flash is also visible but just at the entrance of the pipe.

**Passivated**

Table 1(b) shows an example of a sample with passivated surface finish. Average roughness of 540 nm was measured with a Mitutoyo® profilometer. The sample presented in Table 1(b) exhibits flow marks. This was characteristic of the 8 mm gap, and this behaviour of the electrolyte flow in an annulus system has been observed before in fluid dynamics studies [20][15].

**Reflective and Bright**

A typical reflective and bright sample is presented in Table 1(c). An average roughness of 116 nm was measured with a Mitutoyo® profilometer and 265 nm with an optoelectronic ZYGO® microscope. These values are higher than the expected values of an electrochemical machined surface; this variation however is believed to be due to the curvature of the samples, which made difficult the roughness measurement. A small ring of passivated surface at the exit of the pipe is visible, similar to the passivated entrance – reflective and bright exit samples; this passivation is due to the accumulation of the electrolyte in this area forming a small pool before it can exit the array. The weld-flash is also visible in the entire length of the pipe, and some flow marks can also be observed.

**Reflective and Dark**
A typical example of the samples with reflective and dark surface finish is presented in Table 1(d). Average roughness of 308 nm was measured with a Mitutoyo® profilometer and 426 nm using the optoelectronic ZYGO® microscope. Similar to the results obtained for the reflective and bright samples, this roughness is higher than the expected one due to the curvature of the samples. The same small ring of passivated surface is visible at the exit of these samples, but the weld-flash is barely visible along the length of the pipe. No flow marks can be observed.

Visual similarity between the passivated surface (Table 1(b)) and the reflective and dark surface (Table 1(d)) can be observed; this is due to the presence of the oxide film in both cases. The difference is that in the first case this film is loose, and in the second the film is firmly attached to the surface of the metal.

The ECM parameters, such as current density \( J \), overpotential \( V_0 \) and temperature difference \( \delta T \), which are dependent on the controlled parameters (voltage, gap, electrolyte inlet temperature and flow rate), were compared with the objective of finding which of them had an influence, if any, on the resulting surface finish of the sample. By properly adjusting the process parameters, we expected to achieve a reflective and bright surface finish.

**Overpotential \( V_0 \)**

Metal dissolution was accomplished by the deliberated application of an external potential difference between the electrodes during the ECM process. The potential needed to ensure a continuous discharge of metal between the electrode and the electrolyte is named overpotential \( V_0 \). The overpotential was calculated by using equation (1) [21]:

\[
I = \frac{k_e (V_i - V_0) A}{y}
\]

(1)

where \( k_e \) (S/m) is the electrolyte conductivity, \( y \) (m) is the interelectrode gap, \( V_0 \) (V) is the overpotential, and \( A \) (m²) is the tool surface area.

The results of the present work show the importance of \( V_0 \) in the resulting surface finish at the SS316 samples. Previous works [4, 6, 7] found that a bright and reflective surface finish is achieved when \( V_0 \) is in the range of 9.7 and 12.4 V. In our study, this range is wider, 9.1 – 15.0 V. See Figure 2.
Figure 2. Power, $P$, and overpotential, $V_0$, in relation with the surface finish: passivated entrance – reflective and bright exit (rhomboids), reflective and bright (squares), reflective and dark (triangles), and passivated (circles).

Power supply ($P$) in ECM is a function of the voltage applied and the current measured. $P$ controls the volume and rate of material removed during the process and affects the resulting surface finish. Tailor [22] explained that with a very low combined current-voltage value, the material removal is minimal, and attained this behaviour to the formation of a very thin passive film on the surface of the anode that stops further reactions. At a moderate level of current-voltage combination, brightening of the surface is observed. In this region, the current is high enough to dissolve the material at the surface of the anode. A further increase in voltage generates over-polishing or pitting, and further cause of electrolytic by-products that may generate a dull surface finish [22]. This behaviour is evident in Figure 2 and Figure 3.

The samples that present a more uniform reflective surface finish (squares and triangles in Figure 2) are observed when the power is over 7KW and the overpotential is limited between 7 and 15 V. In the samples where the surface finish is not uniform (rhomboids), i.e. a passivated surface was found at the entrance and a reflective and bright surface at the exit; the overpotential is over 14 V and the power is low (below 7 KW). Published studies [4, 16] have shown that the mass transport control of the dissolved material is a critical factor for the resulting surface finish. Therefore a plausible explanation for this behaviour is that the potential difference is not sufficient to maintain a steady ion migration though the oxide film. Hence a non-uniform surface finish is attained. This is consistent with other studies [8, 16] where the areas of variable surface finish were related with a variation in the valence ($z$) of SS316. In the present work, however it was not possible to measure the valence during the ECM process, and therefore, these findings could not
be verified. Additionally, when the overpotential is below 7 V, the surface finish is passivated (circles in Figure 2). In these cases the potential drop between the electrodes is considered too high. Thus, even though the oxide film is broken, the dissolution of the metal is uncontrolled [23].

**Current density (J)**

Current density, \( J \), can be explained simply as the quantity of electric current that flows on a unit of area during ECM. The current during the ECM process was recorded by a data acquisition software, and the dimensions of the tool/workpiece were known. Hence these data are used to calculate the current density. Figure 3 presents the influence of \( J \) in the resulting surface finish. For \( J \) over 4.5 A/cm\(^2\) (squares and the triangles in Figure 3) a reflective surface is achieved. However, when \( J \) dropped below those values, there was not sufficient energy to break the oxide film uniformly. Hence, a passivated or non-uniform surface finish was obtained. This is in agreement with the work presented by Wang [18] where ECM on stainless steel was carried out under low current density (up to 2.5 A/cm\(^2\)). It was found that at low current density a compact oxide film is formed on the surface of the metal and it can hinder the metal dissolution.

Moreover when \( V_0 \) is too high and \( J \) is not sufficient to maintain a stable migration of ions from the workpiece, a non-uniform surface finish is attained (rhomboids in Figure 3).

![Figure 3](image.png)

**Figure 3** Current density, \( J \), and voltage within the cell, \( V \), in relation with the surface finish: passivated entrance – reflective and bright exit (rhomboids), reflective and bright (squares), reflective and dark (triangles), and passivated (circles).

There is an evident relationship between the overvoltage and the current density. A variation in \( V_0 \) is evidence of a change in the electrochemical reactions at the electrode surface [24] and this affects the
surface finish of the sample. Figure 4 presents this relationship. It can be observed that when $V_0$ increases, more energy is needed to drive the reactions at the sample surface. If this energy is not enough to maintain a stable migration of ions from the workpiece, a non-uniform surface finish is attained (blue rhomboids in Figure 4).

![Figure 4 Current density, $J$, and overpotential, $V_0$, in relation with the surface finish: passivated entrance – reflective and bright exit (rhomboids), reflective and bright (squares), reflective and dark (triangles), and passivated (circles).](image)

In practice, not all the current is used for the removal of material, which influences the results presented here. Other side reactions such as oxide precipitation, or metal dissolution due to high valence, may exist and affect the current density efficiency. Moreover, the electrolyte conductivity plays a crucial role in $J$, $k_e$, in turn, is dependent on the temperature, electrolyte flow rate, electrolyte concentration, etc. [25], thus affecting the overall outcome.

**Flow rate**

The role of the electrolyte flow rate is twofold: it flushes away the metal ions (ECM products) dissolved from the anode before they can reach the cathode and, at the same time, mitigates the temperature increase of the system. The accumulation of the machining products could lead to a shortcut of the system, and the increase of the temperature affects the conductivity of the electrolyte [1]. Hence, there is an impact of the flow rate in the overpotential during the process.

Figure 5 demonstrates the influence of the electrolyte flow rate on the surface finish. Low flow rates ($<3.3 \times 10^{-4} \text{m}^3/\text{s}$) or low electrolyte velocities ($<1 \text{m/s}$) led to the formation of a passive oxide film (which
are subject to pitting) [4] or resulted in a non-uniform surface finish along the sample. This may be due to the heat generated during the ECM process which was not well dissipated. Usually the change in temperature results in a change in the conductivity of the electrolyte, which is temperature dependent, thus affecting the resulting surface finish. McGeough [1] stated that the surface was smoother when the electrolyte velocity was increased. However, if the flow rate is too high (>6.7x10^4 m^3/s) or the electrolyte velocity is over 2 m/s, the process results in a dark surface (triangles in Figure 5). From previous works [26], it is known that this film is mainly formed by Fe, C and small traces of Cr, and that it is strongly attached to the metal surface. Additionally, this oxide film limit the current efficiency [18, 27] during the ECM process. Results show that the flow rate that generates a reflective and bright surface finish is approximately 4.2x10^-4 m^3/s, electrolyte velocities between 1 – 2 m/s (squares in Figure 5 and Figure 6). In addition, high flow rates lead to a small diffusion layer at the anode, hence a higher current density is needed to achieve surface brightening [15]; however, from previous works [28, 29], it was observed that an stable and even flow is critical to keep the supersaturated films (presented by Lohrengel [29]) and to obtain smoother surface finish.

![Figure 5. Electrolyte flow rate, Q , and overpotential, V0 , in relation with the surface finish: passivated entrance – reflective and bright exit (rhomboids), reflective and bright (squares), reflective and dark (triangles), and passivated (circles).](image)

**Interelectrode gap**

The distance between the electrodes and in where the electrolyte flows is named interelectrode gap; and it has been demonstrated to be another important parameter determining the resulting surface finish of the sample. Figure 6 shows that a big gap (≈8 mm) and low electrolyte flow rates (<1 m/s) generates a passivated surface finish. This is due the fact that the interelectrode gap is related inversely to the current.
density of the ECM process [15, 16, 28, 30]; when $J$ is reduced, e.g. by increasing the gap, there is not enough energy during the process to break the oxide film uniformly.

Figure 6 shows the relationship between the surface finish, the electrolyte flow and the interelectrode gap. From fluid dynamic studies, it is known that the velocity of the flow increases when the interelectrode gap is smaller, and that when the electrolyte velocity is too low, the turbulence in the fluid is not sufficient for flushing away of the ECM products, hence affecting the process. This behaviour was also observed by Lee [5], where an optimum gap of 1 mm, in order to obtain a smooth surface, was found.

![Figure 6. Electrolyte velocity and overpotential, $V_o$, in relation with the surface finish: passivated entrance – reflective and bright exit (rhomboids), reflective and bright (squares), reflective and dark (triangles), and passivated (circles).](image)

**Temperature difference ($\delta T$)**

Some samples were found to have two different surface finish along their length following the flow path of the electrolyte. This non-uniform surface finish is usually characterised by a passivated section at the entrance and a reflective and bright one at the exit. We believe that this behaviour is as a result of an increment in temperature of the electrolyte while flowing along the length of the sample due to Joule heating. Hence, for this study the electrolyte inlet temperature for each sample was measured in situ, and the electrolyte outlet temperature was calculated applying heat transfer equations based on the ECM parameters used.

When the temperature increases, the conditions of the ECM process change, resulting in a different surface finish. The change in temperature along the workpiece surface observed in the samples presented here may be a result of a change in the electrolyte flow rate. Hence, the electrolyte cannot control efficiently the
temperature of the electrodes, promoting the change in valence at different points on their surface. It is known that the electric conductivity is directly related with the temperature of the electrolyte [31, 32]. When the conductivity changes, the electrochemical reactions during the ECM also change, thus affecting the resulting surface finish on the sample.

The difference in temperature within the sample may be also affected by the interelectrode gap. ECM products are more easily accumulated in a small gap than in a larger one, equivalently the velocity of the electrolyte flow is higher in the small gap compared with the larger gap. If the electrolyte velocity within the interelectrode gap is not enough, some ECM products may be accumulated at the end of the pipe, provoking a change in the concentration and conductivity of the electrolyte at this point. This effect was also observed by Tang et al. [28] when the surface finish of their samples improved when adding backpressure to the electrolyte flow, hence avoiding the accumulation of products at the surface of the workpiece. For a uniform surface finish, the temperature difference, $\delta T$, of the electrolyte across the length of the pipe, and in consequence its variation in conductivity, should be small (<8 °C).

![Figure 7. Temperature difference, $\delta T$, between the electrolyte inlet and outlet temperature, and overpotential, $V_0$, in relation with the surface finish: passivated entrance – reflective and bright exit (rhomboids), reflective and bright (squares), reflective and dark (triangles), and passivated (circles).](image)

Figure 7 shows $\delta T$ along the length of the samples and their relationship with the resulting surface finish. The samples that present two different surface finish (rhomboids in Figure 7) are the ones whose $\delta T$ is high. This difference is usually result of an electrolyte flow rate that is not enough (<25 L/min) to dissipate
the heat of the ECM process. Additionally, it can be observed that a uniform surface finish is presented when $\delta T$ is low (squares, triangles and circles in Figure 7).

**Valence, $z$**

Valence, $z$, has also been demonstrated to influence the surface finish on SS during the ECM process. The transition from active to transpassive dissolution, and in turn the surface finish, is accompanied by a change in the anode potential, surface microstructure and change in the valence of the metal, as observed in Figure 3 [6, 16]. As it has been seen previously [7, 8, 16], the decrease in dissolution valence corresponds with a progressive change in the electrode surface, i.e. from a smooth bright surface to a passivated one. In the present work the valence during the ECM process could not be measured, so these findings could not be verified for our samples.

**Conclusions**

The paper presented an experimental analysis of the parameters that influence the surface finish of SS316 samples machined with ECM. The machining parameters, voltage, gap, electrolyte flow rate, and electrolyte inlet temperature, were varied in turn, and the samples were divided according to the resulting surface finish.

Results highlighted a strong relationship between surface finish and overpotential during ECM. The overpotential is dependent of the current density and the characteristics of the electrolyte. Thus the electrolyte flow rate, conductivity and inlet temperature directly affect the resulting surface finish. An overpotential between 9 and 15 V is necessary to obtain the desirable reflective and bright surface finish; if the overpotential was lower, a passivated surface usually was obtained. The variation in the electrolyte temperature during the process was found to have a great impact on the uniformity of the surface finish along the sample. A non-uniform surface finish along the length of some samples was characteristic of a low electrolyte flow rate, and hence deficient heat and ECM products dissipation. Additionally the interelectrode gap also affect the resulting surface finish but its influence was not so evident, however a big interelectrode gap, 8 mm, usually results in a passivated surface finish. Current density ($J$) during the ECM process also demonstrated to have a big influence in the resulting surface finish; $J$ higher than 4.5 A/cm$^2$ is needed to obtain a reflective surface finish. However it's important to remember that $J$ is related with the interelectrode gap, and the temperature and conductivity of the electrolyte.

The results presented in this paper could be used as a tool for the achievement of the desired surface finish on a sample of SS316. We believe that this analysis and subsequent further development could be applied with other metal alloys. Moreover, it is an important step towards the understanding of the ECM process and an adequate prediction of the resulting surface finish. The accurate determination of the fundamental relationships between the power, overpotential, interelectrode gap, current density and electrolyte flow rate and electrolyte temperature, will lead to more accurate computational simulations of the ECM process and aid the tool design techniques. However more experimental work is still needed.
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